

Fig. 2. Environments of the ammonium ions viewed along the a axis. The interatomic distances between N and F atoms are given in \hat{A} . E.s.d. = $0.001 \hat{A}$ for all these distances. [Symmetry code: none *x*,*y*,*z*; (i) $x+1$, $y+1$, *z*; (ii) $x+1$, y , *z*; (iii) $x+1$, $y-1$, *z*; (iv) *x*, 1 + y, z; (v) $\frac{1}{2}$ + x, $\frac{3}{2}$ - y, z - $\frac{1}{2}$; (vi) x, y - 1, z; (vii) $\frac{1}{2}$ + x, $\frac{1}{2}$ - y, $\frac{1}{2}$ + z.]

In spite of satisfactory R values the standard deviations of the atomic positions are high which suggests some disorder in the structure. At room temperature strong diffuse streaks parallel to the b^* axis were observed on Weissenberg photographs. This diffuse scattering may be connected with the partial disorder in the structure and is probably related to the mechanism of the phase transition. At 334.4 K the crystal passes into the orthorhombic phase and in place of the diffuse streaks there appear quite new and sharp Bragg reflections which are consistent with the space group $P2_12_12_1$. The crystal structure of this phase and measurements of the lattice parameters as a function of temperature will be the subject of a subsequent paper.

The author would like to thank Dr Z. Czapla from the University of Wrocław for his kind provision of the crystals, Professor K. Lukaszewicz for stimulating discussions and Dr U. Rychlewska from the A. Mickiewicz University of Poznań for collecting the diffractometer data.

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Acta Cryst. (1983). C39, 1169-1172

Structure of Orthopyroxene-Type and Clinopyroxene-Type Magnesium Germanium Oxide MgGeO₃

BY MITUKO OZIMA

Institute for Solid State Physics, University of Tokyo, Tokyo 106, *Japan*

(Received 2 February 1983; *accepted 7 June* 1983)

Abstract. (i) Orthopyroxene-type: $M_r = 144.92$, orthorhombic, *Pbca*, $a = 18.8099$ (12), $b = 8.9484$ (8), $c=5.3451(4)~\text{\AA}$, $V=899.69(12)~\text{\AA}^3$, $Z=16$, $D_x=$ 4.28 gcm⁻³, $\lambda(Ag K\alpha) = 0.56087 \text{ Å}, \mu(Ag K\alpha) =$ 75.64 cm⁻¹, $T = 300$ K, $F(000) = 1088$, $R = 0.025$ for 975 independent reflections. (ii) Clinopyroxene-type: $M_r = 144.92$, monoclinic, $C2/c$, $a = 9.6010(8)$, $b =$ 8.9323 (6), $c = 5.1592$ (5) Å, $\beta = 101.034$ (9)°, $V =$ 434.27 (6) Å³, $Z = 8$, $D_x = 4.43$ gcm⁻³, $\lambda(Ag K\alpha) =$ $0.56087 \text{ Å}, \quad \mu(\text{Ag } K\alpha) = 78.35 \text{ cm}^{-1}, \quad T = 300 \text{ K},$ $F(000) = 544$, $R = 0.021$ for 1025 independent reflec-

tions. Orthorhombic $MgGeO₃$ (the high-temperature and low-pressure phase) is isostructural with orthoenstatite, MgSiO₃. The structure of monoclinic $MgGeO$, (the low-temperature and high-pressure phase) is different from that of clinoenstatite which crystallizes in *P21/c.* The orthorhombic *(Pbca)* to monoclinic $(C2/c)$ transformation in MgGeO₃ is accompanied by a fairly large volume decrease (3.5%) . A smaller distortion of the Mg(2) octahedron in the structure of monoclinic $MgGeO₃$ causes the higher density.

0108-2701/83/091169-04501.50

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Introduction. Enstatite, MgSiO₃, one of the most important pyroxene-group minerals, is known to show a complicated polymorphism. Single crystals of orthoenstatite have successfully been grown by Ito (1975) and Ozima (1982) using a flux method. However, growth of single crystals of untwinned clinoenstatite has never been reported, and the structure analysis of the clinoenstatite has been carried out on the 'ortho-inverted-clinoenstatite' which contains very fine polysynthetic twinning (Morimoto, Appleman & Evans, 1960).

Germanate pyroxene, $MgGeO₃$, is frequently used as a convenient high-pressure model substance for the corresponding silicate, enstatite, on account of its lower transition pressure. Recently, Ozima & Akimoto (1983) have demonstrated the successful growth of single crystals of $MgGeO₃$ with the orthopyroxene-type structure as well as the clinopyroxene-type structure at atmospheric pressure by the flux method. They also established that the orthorhombic $MgGeO₃$ is the high-temperature and low-pressure phase and that monoclinic $MgGeO₃$ is the low-temperature and highpressure phase. This phase relation is very similar to that of enstatite but completely different from that reported previously by Kirfel & Neuhaus (1974).

In this paper, the crystal structures of both orthorhombic and untwinned monoclinic $MgGeO₃$ are reported together with discussion of the transition mechanism of the orthorhombic phase to monoclinic phase in $MgGeO₃$.

Experimental. (i) Orthopyroxene-type: An octagonal columnar crystal formed by $\{210\}$, $\{100\}$ and $\{010\}$ of volume 1.0×10^{-6} cm³; Rigaku automated four-circle diffractometer, graphite-monochromated Ag $K\alpha$ radiation; unit-cell parameters from a least-squares fit for 30 reflections with $30^{\circ} \le 2\theta \le 45^{\circ}$; integrated intensities measured by $2\theta-\omega$ scans; 1197 reflections measured, 975 independent reflections ($|F_q| > 2\sigma |F_q|$) within 0° <2 $\theta \le 45^{\circ}$, 222 unobserved, $(\sin \theta/\lambda)_{\text{max}} =$ 0.682 Å^{-1} , $0 \le h \le 25$, $0 \le k \le 12$, $0 \le l \le 7$; three standard reflections every fifty reflections; corrections for Lorentz-polarization effects and absorption (transmission factor ranged from 0.52 to 0.60 ; heavy atommethod, full matrix least-squares refinement on F using *RADIEL* (Coppens, Guru-Row, Leung, Stevens, Becker & Yang, 1979), unit weights, anisotropic thermal parameters for all atoms; refinements of the formal charges of the Ge and the bridging oxygen atoms were carried out; fully-ionized scattering factors for Ge^{4+} and Mg^{2+} , orbital scattering factor for Ge, and the atomic scattering factor for O from *International Tables for X-ray Crystallography* (1974); the scattering factor for Q^{2-} from Tokonami (1965); isotropic secondary extinction effect assumed [refined G factor $0.61(2) \times 10^{-4}$; $R = 0.025$, $wR(F) = 0.026$, $S =$

1.4474; final $\Delta/\sigma = 0.06$, Δp in final difference map $= -0.66 - 0.90$ e Å⁻³.

(ii) Clinopyroxene-type: A tetragonal columnar crystal formed by $\{110\}$ of volume 7.8×10^{-7} cm³, confirmed to be not twinned by precession photographs in the *a*-c** plane; unit-cell parameters obtained from a least-squares fit for 32 reflections with $30^{\circ} \le 2\theta \le 45^{\circ}$; 1271 reflections measured, 1025 independent reflections $(|F_{\alpha}| > 2\sigma|F_{\alpha}|)$ within $0^{\circ} < 2\theta \le 60^{\circ}$, 192 unobserved, $(\sin \theta/\lambda)_{\text{max}} = 0.891A^{-1}, \quad -17 \le h \le 17, \quad 0 \le k \le 15,$ $0 \le l \le 9$; transmission factor ranged 0.58 to 0.67; refined G factor 7.6 (2) \times 10⁻⁴; $R = 0.021$, $wR(F)$ $= 0.026$, $S = 1.9904$; final $\Delta/\sigma = 0.02$, $\Delta\rho$ in final difference map $= -0.44 - 0.72$ e Å⁻³.

Discussion. Final atomic parameters are given in Table l.t The *OR TEP* plot of the structures of the orthorhombic and monoclinic phases in $MgGeO₃$ is shown in Fig. 1 (Johnson, 1965). Both belong to the pyroxene structure, *i.e.* GeO₃ chains, along the c axis form the framework of the crystal. In Table 2, bond distances and bond angles are listed. It is observed that the distances between Ge and the bridging O atoms are longer than those between Ge and other O atoms. This seems to be related to a covalent character of the Ge-bridging-oxygen bond. (The valences of Ge and the bridging O atoms were found to be about $+3$ and -1 , respectively.)

t Lists of structure factors, anisotropic thermal parameters and Table 3 have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 38645 (19 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

Table 1. *Fractional coordinates* $(x 10⁵)$ *and equivalent isotropic thermal parameters* (\times 10⁴) *with e.s.d.'s in parentheses for orthorhombic and monoclinic* MgGeO₃

Table 2 *(cont.)*

α and notified wight α_3		
$Ge(1) - O(1A)$ $-O(2A)$ $-O(3A^{\nu})$ $-O(3A^{iv})$ $O(1A) - O(2A)$ $-O(3A^{iv})$ $-O(3A^{\nu})$ $O(2A) - O(3A^{iv})$ $-O(3A^{\nu})$ $O(3A^v) - O(3A^iv)$	1.739(3) 1.708(3) 1.795(3) 1.799(3) 2.960(4) 2.853(4) 2.807(4) 2.787(4) 2.875(4) 2.951(6)	118.30(14) 107.45(14) $105 \cdot 16(14)$ 105.20(14) 110.27(14) 110.39(14)
$Ge(2) - O(1B)$ $-O(2B)$ $-O(3B)$ $-O(3B^{\text{viii}})$ $O(1B) - O(2B)$ $-O(3B)$ $-O(3Bviii)$ $O(2B) - O(3B)$ $-O(3Bviii)$ $O(3B) - O(3B^{\text{viii}})$	1.725(3) 1.705(3) 1.787(3) 1.762(3) 2.948(4) 2.962(4) 2.840(4) 2.586(4) 2.919(4) 2.765(6)	118.52(14) 114.99(14) 109.08(14) 95.55(14) 114.69(14) 102.37(14)
$Mg(1) - O(1A^{ii})$ $-O(1A^{iii})$ $-O(1B^{vi})$ $-O(1B^{\text{vii}})$ $-O(2A)$ $-O(2B^{\nu})$ $O(1A^{ii}) - O(2A)$ $-O(1B^{\rm vi})$ $-O(1B^{\text{vil}})$ $O(1A^{ i}) - O(1A^{ })$ $-O(2A)$ $-O(1B^{\text{vii}})$ $-O(2B^{\nu})$ $O(2A) - O(1B^{v})$ $-O(2B^{\nu})$ $O(1B^{vi}) - O(1B^{vii})$ $-O(2B^{\nu})$ $O(1B^{vii})-O(2B^{v})$	2.161(3) 2.097(3) 2.055(3) 2.154(3) 2.055(3) 2.004(3) $3-131(4)$ 2.825(4) 2.811(4) 3.107(6) 3.157(4) 2.825(4) 2.783(4) 2.693(4) 2.961(4) 3.122(6) 3.031(4) 2.919(4)	95.89(13) 84.13(12) 81.32(12) 93.72(13) 99.01(13) 83.29(13) 85.42(13) 81.88(13) 93.66(14) 95.73(13) 96.60(13) 89.11(13)
$Mg(2) - O(1A^{i})$ $-O(1B^{vi})$ $-O(2A)$ $-O(2B)$ $-O(3A)$ $-O(3B)$ $O(1A^{i})-O(2A)$ $-O(3A)$ $-O(1B^{vi})$ $-O(2B)$ $O(2A) - O(3A)$ $-O(1B^{v_i})$ $-O(3B)$ $O(3A) - O(2B)$ $-O(3B)$ $O(2B) - O(1B^{V})$ $-O(3B)$ $O(3B) - O(1B^{vi})$	2.067(3) 2.100(3) 2.009(3) 2.078(4) 2.286(3) 2.228(3) 3.019(4) 3.078(4) 2.811(4) 2.783(4) 3.293(4) 2.693(4) 3.433(4) 3.074(4) 2.841(4) 2.922(4) 2.586(4) 3.467(4)	95.54(14) 89.85(13) 84.85(13) 84.34(13) 99.94 (13) 81.87(13) $108 \cdot 10(13)$ 89.46 (13) 78.01(12) 88.76(13) 73.74(13) $106-41(13)$

Symmetry code: (i) $-x$, $1-y$, $1-z$; (ii) $-x$, $\frac{1}{2}+y$, $1\frac{1}{2}-z$; (iii) $-x$, $1-y$, 2-z; (iv) x, $\frac{1}{2}$ -y, $\frac{1}{2}$ +z; (v) x, y, 1+z; (vi) $\frac{1}{2}$ -x, 1-y, $\frac{1}{2}$ +z; (vii) $\frac{1}{2}$ -x, $\frac{1}{2}+y$, $1+z$; (viii) $x, \frac{1}{2}-y, -\frac{1}{2}+z$.

Symmetry code: (i) *x*, $1-y$, $\frac{1}{2}+z$; (ii) $1-x$, y , $\frac{1}{2}-z$; (iii) $1-x$, y , $1\frac{1}{2}-z$; (iv) x, y, $-1+z$; (v) $\frac{1}{2}-x$, $\frac{1}{2}-y$, $1-z$; (vi) $\frac{1}{2}-x$, $-\frac{1}{2}+y$, $1\frac{1}{2}-z$; (vii) $\frac{1}{2} + x$, $-\frac{1}{2} + y$, z; (viii) $\frac{1}{2} + x$, $\frac{1}{2} - y$, $-\frac{1}{2} + z$; (ix) $\frac{1}{2} + x$, $\frac{1}{2} - y$, $\frac{1}{2} + z$; (x) x, $1-y$, $-\frac{1}{2}+z$.

 $[O(3B^{viii}) – O(3B) – O(3B^{iv})$ angle = 150.25 (15)^o] than the other $[O(3A)-O(3A^{iv})-O(3A^{v})$ angle = 129.79 (14)°]. This structure is identical with that of orthoenstatite, MgSiO₃ (Morimoto & Koto, 1969). The structure of the monoclinic phase of $MgGeO₃$, however, is different from that of clinoenstatite, $MgSiO₃$ (Morimoto, Appleman & Evans, 1960). Clinoenstatite, possesses two kinds of GeO_3 chains, similar to orthoenstatite, and the space group is $P2₁/c$, while the monoclinic phase of $MgGeO$, possesses only one kind of $GeO₃$ chain, which corresponds to the more kinked chain in orthorhombic $MgGeO_3$ $[O(2^i)-O(2)-O(2^x)]$ angle = $129.20(12)$ and the space group is $C2/c$. Thus the structure of the monoclinic phase of $MgGeO₃$ is similar to that of diopside $(CaMgSi₂O₆)$. As indicated in Table 2, for both orthorhombic and monoclinic $MgGeO₃$, the octahedral coordination of O atoms around $Mg(1)$ is regular, while the bridging oxygen $[O(3B)]$ of the less kinked chain $[Ge(2)]$ tetrahedron] causes the large distortion of the Mg (2)

$MgGeO₃$

Fig. 1. Stereoscopic projection along the b axis of the structure of (i) orthorhombic and (ii) monoclinic $MgGeO₃$. Atoms are represented by their thermal ellipsoids with a probability of 50%.

octahedron in orthorhombic $MgGeO₃$. Therefore, the distortion of the Mg(2) octahedron in monoclinic $MgGeO₃$ is smaller than in orthorhombic $MgGeO₃$. One could call the structure of monoclinic $MgGeO₃$ 'simpler' than that of orthorhombic $MgGeO₃$.

In Table 3,* crystal data of silicate and germanate pyroxenes are compiled and the change in the unit-cell volume during the orthorhombic \rightarrow monoclinic transformation is shown. Morimoto, Nakajima, Syono, Akimoto & Matsui (1975) have pointed out that the polymorphism relation in pyroxenes is only observed when the orthorhombic form is in space group *Pbca* and the monoclinic in space group $P2₁/c$. No monoclinic pyroxene in space group *C2/c* has ever been reported to have orthorhombic polymorphs *(Pbca)* except $ZnSiO₃$. As shown in Table 3, their observation

holds true only for silicate pyroxenes, not for germanates. $ZnSiO₃$ is exceptional and cannot be compared with the other pyroxenes, because $\mathbb{Z}n^{2+}$ in the $M(2)$ site has fourfold coordination (Morimoto, Nakajima, Syono, Akimoto & Matsui, 1975). It should be pointed out that the polymorphism between the orthopyroxene structure *(Pbca)* and the clinopyroxene structure *(C2/c)* in the germanate pyroxene induces a fairly large change in cell volume $(-3.5\%$ and -2.2% in MgGeO₃ and $CoGeO₃$, respectively). On the other hand, in the case of silicate pyroxene, the volume change is practically nil $(+0.2 \sim -0.2\%)$. This suggests that the pyroxene structure with *C2/c* is more compact than that with $P2₁/c$. The large volume change could be explained by the smaller distortion of the $Mg(2)$ octahedron in the germanate clinopyroxene in space group *C2/c.*

When Figs. $1(i)$ and $1(ii)$ are compared, the structural relations of the orthorhombic MgGeO, and monoclinic $MgGeO₃$ can be interpreted as follows. If the individual $Ge(2)$ tetrahedron in orthorhombic $MgGeO₃$ rotates around the a axis, the Ge(2) chain becomes like the Ge(1) chain, increasing the shrinkage of the chain. Furthermore, if this shrunken Ge(2) chain slips about $\frac{1}{4}$ along the c axis, this structure becomes just like that of monoclinic $MgGeO₃$. During these operations, the $Mg(2)$ -O(3B) bond is broken and transferred to $Mg(2)$ -O(3B^{iv}), all other bonds being maintained. This orthorhombic-monoclinic transformation is, therefore, not martensitic.

I am very grateful to Professor S. Akimoto and Dr M. Konno of the Institute for Solid State Physics, University of Tokyo, for their help and discussions.

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^{*} Deposited; see deposition footnote.